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Synthesis, structure and thermogravimetric analysis of 2 α, ω -telechelic polydimethylsiloxanes of low molecular weight

ALEKSANDRA M. TASIĆ¹, MARIJA V. PERGAL^{2#}, MALIŠA P. ANTIĆ³ and 3 4 VESNA V. ANTIĆ3* 5 ¹Scientific Veterinary Institute of Serbia, Vojvode Toze 14, Belgrade 11000, Serbia, ²Institute of Chemistry, Technology and Metallurgy, University of Belgrade, Njegoševa 6 12, Belgrade 11000, Serbia and ³University of Belgrade, Faculty of Agriculture, 7 8 Nemanjina 6, Zemun, Belgrade 11080, Serbia 9 (Received 27 April, revised 19 June, accepted 28 June 2017) 10 Abstract: A series of α, ω -telechelic polydimethylsiloxanes (PDMS), with predetermined molecular weights of about 2500 g mol-1, was synthesized by 11 12 siloxane equilibration reaction. Syntheses were performed using octamethyl-13 cyclotetrasiloxane (D_4) and various disiloxanes: hexamethyldisiloxane (HMDS), 14 1,1,3,3-tetramethyldisiloxane (TMDS), 1,3-divinyltetramethyldisiloxane 15 (DVTMDS), 1,3-bis(3-carboxypropyl)tetramethyldisiloxane (DCPTMDS) and 16 1,3-bis(3-aminopropyl)tetramethyldisiloxane (DAPTMDS). The role of the 17 disiloxane was to introduce terminal functional groups at the end of the 18 polymer chains and to control the molecular weight of the polymers. Polymers 19 with trimethyl, hydrido, vinyl, carboxypropyl and aminopropyl end-groups were obtained in this way. The structure of the α, ω -telechelic PDMSs was 20 21 22 23 24 confirmed by NMR and IR spectroscopy. The molecular weights of the polymers were determined by ¹H-NMR, gel permeation chromatography (GPC) and dilute solution viscometry. Thermogravimetric analysis (TGA) under nitrogen and air showed that the type of the terminal groups significantly 25 influenced the thermal and thermo-oxidative stability, as well as the 26 degradation mechanism of the α, ω -telechelic PDMSs. 27 Keywords: a, a-telechelic PDMS; siloxane equilibration; thermal and thermo-28 oxidative degradation. 29 INTRODUCTION

30 Linear polydimethylsiloxanes (PDMS) with potentially reactive functional groups at their chain ends, known as α, ω -telechelic PDMS, can be synthesized 31 by the so-called equilibration reactions of functional disiloxanes and cyclic silox-32

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anes in the presence of an appropriate reaction catalyst. The general synthetic

Functional disiloxanes (DS in Fig. 1) are the key starting reactants for the 37 synthesis of α, ω -telechelic PDMS, since they not only determine the type of the 38 39 end-groups, but also play the role of molecular weight regulator. The most com-40 monly used cyclic siloxane is octamethylcyclotetrasiloxane (D_4 in Fig. 1). The 41 catalyst can be an acid or base compound, which breaks siloxane bonds in linear 42 and cyclic siloxanes, while the Si-CH₂, Si-R and R-X bonds remain intact. Group X in Fig. 1 is a reactive functional group until R is short organic moiety, 43 44 which can or cannot exist. α, ω -Telechelic PDMSs, that possess reactive X group 45 bounded over moiety R, show several advantages in comparison with their Si-X 46 analogs. First, the type and nature of the residue R has a significant effect on the 47 chemical reactivity of the functional group X. Moreover, a suitable choice of the 48 residue R may affect the solubility of Si-R-X terminated siloxanes, as well as 49 their miscibility with appropriate organic monomers or polymers.¹ Strictly 50 speaking, X does not include CH₃ groups, which are seen in "siloxane oil". 51 However, polysiloxanes with terminal methyl groups are considered as telechelic 52 PDMSs, since they often play the role of reference polymers.

53 α, ω -Telechelic PDMSs of low molecular weights are versatile starting mat-54 erials for the preparation of a wide variety of linear siloxane-containing copoly-55 mers, such as siloxane-urethanes, siloxane-amides, siloxane-imides and silox-56 ane-esters, which possess properties of thermoplastic elastomers. α, ω -Telechelic 57 PDMSs are also used in the modification of network structures. Interest in 58 PDMSs as reactive prepolymers is due to their unique combination of properties, 59 which include extremely low glass transition temperature (-123 °C), excellent

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thermal, oxidative and UV stability, very low surface energy, high gas permeability and biocompatibility (physiological inertness). Due to the mentioned properties, organic–siloxane copolymers have received special attention as elastomers, protective coatings, photoresists, biomaterials, gas separation membranes,
emulsifiers, *etc.*^{1–3}

In the hitherto published papers, homogeneous catalysts, such as sulfuric, 65 66 trifluoromethanesulfonic and trifluoroacetic acid (acid type), or potassium and tetramethylammonium hydroxide (base type), were mostly used for the prepar-67 ation of α, ω -telechelic PDMS, 4-6 with very few exceptions. On the other hand, it 68 69 has long been known that siloxane equilibration polymerization can also be cat-70 alyzed by catalysts which are insoluble in the respective reaction mixture, such as, for example, ion-exchange resins.^{7–12} The resulting heterogeneous catalysts 71 exhibit many practical advantages over the homogeneous ones.^{13,14} For example, 72 73 a heterogeneously-catalyzed equilibration polymerization does not require time 74 consuming and sometimes very tedious post-polymerization procedures, 75 including catalyst deactivation, neutralization and isolation, which are necessary 76 in homogeneous catalysis. Generally, these catalysts can be recuperated, 77 permitting their multiple usages after simple separation, washing and drying. 78 Among many different types of insoluble catalysts, particular attention has been 79 devoted to cation-exchange resins (CER), since they are cheap, thermally stable, easy to handle and commercially available.¹⁵ 80

Considering the importance of α, ω -telechelic PDMSs, a relatively small 81 82 number of studies dealing with the impact of the terminal groups on polymer 83 thermal and thermo-oxidative stability have been published to date.^{16–21} It is 84 well known that pronounced stability of polysiloxanes at elevated temperatures is 85 a direct consequence of their fundamental properties, such as inherent strength of 86 the Si-O bond, outstanding flexibility of the $-[Si-O]_x$ - chain segments and 87 higher entropic stability of the low molecular weight cyclic siloxanes than of the linear siloxane polymers at degradation temperatures.^{2,3,22-25} Polysiloxanes as a 88 class are thermally more stable than their -C-C- counterparts. The temperature 89 90 of the onset of irreversible thermal degradation of PDMS, the parent polymer of the polysiloxane family, may reach 300-400 °C^{2,3} that is much higher than the 91 92 degradation temperatures of most organic -C-C- polymers, the stability of which 93 rarely exceeds 150-200 °C. The differences in degradation behavior of PDMSs 94 result from three main factors: 1) the degradation atmosphere, 2) the type and the 95 relative concentration of polymer end-groups²⁶ and 3) the purity of the examined 96 sample, *i.e.*, the presence or absence of even catalytic amounts of ionic impur-97 ities, including additives or residual polymerization catalyst.^{3,24} These factors 98 can completely change the mechanism of the degradation process and hence the 99 characteristic features of PDMS polymers. As a consequence, in inert atmo-100 sphere, PDMSs may degrade by three different reaction mechanisms, including:

101 1) the "unzipping" mechanism (proposed by Alexandrova and Rode in 1968-102 -1969),^{27,28} 2) the "random scission" mechanism (proposed by Thomas and Kendrick in 1969)²⁹ and 3) the "externally catalyzed" mechanism (proposed by 103 Grassie and Macfarlane in 1978).³⁰ In dynamic thermal gravimetric analysis, in 104 105 the absence of oxygen, PDMSs generally degrade in a single weight-loss step. On 106 the other hand, thermo-oxidative degradation of the same PDMS is generally a 107 more complex process, which usually shows two weight-loss steps, leading to 108 pure silica at temperatures above 600 °C.

109 In this work, heterogeneous and homogeneous catalysis was combined for 110 the preparation of α, ω -telechelic PDMSs, with targeted molecular weights of 111 about 2500 g mol⁻¹. A cation-exchange resin based on macroporous sulfonated 112 crosslinked polystyrene was used as the heterogeneous catalyst for the synthesis 113 of trimethyl, hydrido, vinyl, and carboxypropyl terminated PDMS, while PDMS 114 with aminopropyl end-groups was prepared using tetramethylammonium hyd-115 roxide as a homogeneous catalyst. Polymers of low molecular weight were 116 obtained in all cases, in order to be later used for the synthesis of block copoly-117 mers. The syntheses were performed starting from D₄, while the disiloxane co-118 reactants for functional group incorporation were hexamethyldisiloxane (HMDS), 119 1,3-divinyltetramethyldisiloxane 1,1,3,3-tetramethyldisiloxane (TMDS), 120 (DVTMDS), 1,3-bis(3-carboxypropyl)tetramethyldisiloxane (DCPTMDS) and 121 1,3-bis(3-aminopropyl)tetramethyldisiloxane (DAPTMDS). The structures of the 122 α, ω -telechelic PDMSs were examined by NMR and IR spectroscopy. Molecular weights were obtained by ¹H-NMR spectroscopy, gel-permeation chromato-123 124 graphy (GPC) and dilute solution viscometry. Thermogravimetric analysis in nit-125 rogen and air was used to examine the influence of the end-groups on thermal and 126 thermo-oxidative stability of α, ω -telechelic PDMSs. Commercial PDMS samples 127 with terminal methyl and silanol groups were also analyzed as reference polymers.

EXPERIMENTAL

129 Materials

128

D₄ (98.9 %), HMDS (100 %), TMDS (82.7 %), DVTMDS (98.7 %), DCPTMDS and 130 131 DAPTMDS (61.8 %), are purchased from ABCR (Germany). In parentheses are shown the 132 purities of the liquid chemicals, which were determined by gas chromatography, while the 133 molecular structure was confirmed by NMR spectroscopy. DAPTMDS was also analyzed by 134 IR spectroscopy.

135 A commercial grade macroporous cation-exchange resin Duolite C26 (CER), strong acid 136 type, from Diamond Shamrock (USA), having a total ion-exchange acid capacity of 1.85 eq L⁻¹, 137 was used as the reaction catalyst after overnight drying by warming at 50 °C in a vacuum 138 dryer. An aqueous solution of tetramethylammonium hydroxide, TMAH, concentration of 25 139 %, purchased from Fisher Scientific (England), was used as received. Chloroform from Lach-140 ema (Czech Republic), was distilled at atmospheric pressure, and the fraction boiling at 61 °C 141 was collected. Toluene, from Alkaloid-Skopje, was distilled at atmospheric pressure, and the 142 fraction boiling at 110 °C was collected.

143 Constant viscosity PDMS standards (B1-B7, Brookfield Laboratories, USA), terminated 144 entirely with methyl groups were used for calibration of the GPC system. Cyclosiloxanes, D₄, 145 and decamethylcyclopentasiloxane, D5 (ABCR, Germany) were also used for the construction 146 of a calibration curve. The average molecular weights of the Brookfield standards were cal-147 culated from their zero shear viscosities.³¹ The zero shear viscosities and the molecular weights 148 of the Brookfield standards are given in Table S-I of the Supplementary material. The stan-149 dard sample B3, with η_0 of 50.5 cSt and molecular weight of 3270 g mol⁻¹ (Table S-I of the 150 Supplementary material to this paper), was used for thermogravimetric analysis, as well as a 151 commercial a, a-disilanol-polydimethylsiloxane, PDMS-OH (ABCR, Germany). The mole-152 cular mass of the sample PDMS-OH, according to the producer specification, was 2000-3500 153 g mol⁻¹, with η_0 of 45–85 cSt.

154 Synthesis of α,ω-telechelic PDMSs

All siloxane equilibration reactions were performed in a 100 cm³, four-necked, round--bottomed flask equipped with a mechanical stirrer, reflux condenser, gas inlet and a thermometer. The compositions of the reaction mixtures are given in Table I.

158 In syntheses No. 1, 3 and 4, the appropriate amount of D_4 , disiloxane (DS) and the cat-159 alyst (CER) were charged into the reactor flask, and the flask was placed in a constant tempe-160 rature oil bath at 50 °C. The amount of CER catalyst was in all cases 22 meg per 100 g of 161 reaction mixture. The reaction mixture was stirred for 2 h at 50 °C, and then 22 h at 95 °C. 162 After stirring the reaction mixture for 24 h, at a speed of 500 rpm, it was filtered through a 163 stainless steel net filter to remove the catalyst. The filtrate was fractionated by distillation 164 under a reduced pressure of 0.5 mm Hg (66.7 Pa) to remove the volatiles. The equilibrate was 165 gradually heated to a temperature of 200 °C and then this temperature held constant for 30 166 min. From the weight of the separated fractions of cyclics and linear polymers, the 167 quantitative composition of equilibrates was calculated. The catalyst was washed with three 168 portions of dichloromethane and then dried 24 h at 50 °C before the next use.

Synthesis No. 2 (sample PDMS-H) was performed with a different temperature regime.
Due to the relatively low boiling temperature of TMDS (70 °C), the reaction mixture was first
thermostated at 50 °C for 22 h, then 1 h at 65 °C and finally 1 h at 95 °C. The remaining
procedure was completely the same as previously described.

173 Synthesis No. 5 was performed using tetramethylammonium hydroxide as a homogen-174 eous catalyst, in the following way: appropriate amounts of D_4 , DAPTMDS and TMAH (0.5 g 175 of TMAH per 100 g of reaction mixture) were charged into the reaction flask and the mixture 176 was stirred 2 h at 50 °C, then 22 h at 95 °C and finally 1 h at 150 °C. Heating at 150 °C 177 provokes thermal degradation of TMAH catalyst. The remaining procedure was completely 178 the same as previously described for the polymers obtained with CER catalyst.

179 Measurements

180 *NMR spectroscopy.* ¹H- (200 MHz), ¹³C- (50 MHz) and DEPT-NMR spectra were 181 obtained on a Varian Gemini-200 instrument. The samples were solutions in CDCl₃, with tet-182 ramethylsilane (TMS) as the internal standard.

183 Gas chromatography, GC. Gas chromatography was realized on a Hewlett Packard, model
 184 HP 5890S instrument, with a 4 m long DB-wax column and a flame ionization detector. The
 185 injector temperature was 250 °C, the column was held 3 min at 50 °C and then heated at 10 °C
 186 min⁻¹ to 270 °C, and the detector temperature was 300 °C.

Infrared spectroscopy, IR. A Perkin–Elmer Model 1725X spectrophotometer was used to
 record the IR spectra of the DAPTMDS and PDMS-NH₂ samples as films on KBr pellets.

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189 Intrinsic viscosometry. The intrinsic viscosities, $[\eta]$, of the PDMS samples were calcul-190 ated based on the flow time, t, in toluene at 25 °C, using an Ubbelohde viscosimeter (l = 86.05191 mm, R = 1 mm). The concentrations of the samples were 0.2–2.0 g dL⁻¹. The mean value of 192 two measurements of t was taken to calculate the relative viscosity, $\eta_r = t/t_0$, where t_0 is the 193 flow time of the solvent. Based on the relative viscosity, the specific viscosity (η_{sp}) was cal-194 culated as $\eta_{sp} = \eta_r - 1$. 195

The reduced, η_{red} , and inherent, η_{inh} , viscosities were calculated by Eq (1) and (2):

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$$\eta_{\rm red} = \eta_{\rm sp}/c \tag{1}$$

$$\eta_{\rm inh} = \ln \eta_{\rm r}/c \tag{2}$$

(3)

198 where c is concentration, in g dL⁻¹.

199 Extrapolation of the straight-lines of η_{red} and η_{inh} , as functions of solution concentration 200 to zero concentration gave [η]. The mean viscometric molecular weight (M_{η}) was calculated 201 using the Mark-Houwink equation:

$$[\eta] = K \overline{M}_{\eta}^{a}$$

where the values of $K = 2.15 \times 10^{-4}$ dL g⁻¹ and a = 0.65 are constants for the system 203 PDMS/toluene at 25 °C.32 204

205 Gel permeation chromatography, GPC. Gel permeation chromatography (GPC) was 206 performed on a Waters 600E instrument equipped with a refractive index detector, on three 207 Supelco Pl-Gel columns connected in line (cross-linked polystyrene with pore sizes of 10^{-5} . 208 10⁻⁶ and 10⁻⁷ m, respectively) at 30 °C. Twenty microliters of a 10 wt.% sample in chloroform 209 solution was injected in all cases. The flow-rate of the chloroform was 1.5 cm³ min⁻¹. The sys-210 tem was calibrated with Brookfield PDMS standards (Table S-I), D₄ and D₅. Elution volumes (V_e) were determined and the calibration curve $V_e = f(\log M)$ was constructed, where \overline{M} is 211 212 the average molecular weight calculated based on the Barry's equation.³¹ The calibration 213 curve is depicted in Fig. S-1 of the Supplementary material.

214 Thermogravimetric analysis, TGA. The thermal and thermo-oxidative stability of the 215 PDMS samples were determined using a TA Instruments SDT Q600 analyzer in the tempe-216 rature range from 30 to 700 °C. The heating rate was 10 °C min⁻¹. The TGA scans were rec-217orded under a dynamic nitrogen or air atmosphere with gas flow rates of 100 cm³ min⁻¹. The 218 sample mass was around 10 mg,

RESULTS AND DISCUSSION

220 Synthesis and structure of α , ω -telechelic PDMSs

221 Four heterogeneously and one homogeneously catalyzed siloxane equilib-222 ration reaction are performed to obtain α, ω -telechelic PDMSs with targeted 223 molecular weight of 2500 g mol⁻¹. The syntheses were performed starting from 224 D₄ and the required disiloxane. The reaction scheme is shown in Fig. 1.

225 The structural formulas of disiloxanes, the role of which was the introduct-226 ion of functional groups and molecular weight regulation, as well as the formulas 227 of α, ω -telechelic PDMSs, are shown in Fig. 2. The more correct titles of the end-228 -groups of the polymers would be (trimethylsilyl)oxy (PDMS-CH₃), (dimethyl-229 silyl)oxy (PDMS-H), (dimethylvinylsilyl)oxy (PDMS-CH=CH₂), [(3-230 carboxypropyl)dimethylsilyl]oxy (PDMS-COOH) and [(3-aminopropyl)

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dimethylsilyloxy (PDMS-NH₂). As the siloxy residue exists in all structures
shown, simplified titles were assigned to the polymers and are used in Fig. 2.



 \sum

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Comment [V5]: Please correct: 1) All " α, ω -" should be in *italic* Changes for α, ω -Telechelic polymers nomenclatu a) α, ω -bis(trimethyl)polydimethylsiloxane ("bis" should not be in italic) b) α, ω -dihydropolydimethylsiloxane c) α, ω -divinylpolydimethylsiloxane d) α, ω -bis(3-carboxypropyl)polydimethylsiloxane e) α, ω -bis(3-aminopropyl)polydimethylsiloxane



In heterogeneously catalyzed reactions, where HMDS, TMDS, DVTMDS and DCPTMDS were used as the disiloxane, the catalyst was CER, while in homogeneously catalyzed reaction, where DAPTMDS was used as the disiloxane, the catalyst was TMAH. The synthesis of α, ω -diaminopropyl PDMS could not be realized with an acid catalyst such as CER, because the amino groups of DAPTMDS are strong proton-acceptors. The protons, which catalyze siloxane redistribution reaction, are "caught" by the free electron pairs of the nitrogen of

242 amino groups. As shown in Fig. 3, salt formation occurs in the presence of DAPTMDS, which leads to inactivation of the CER catalyst.³³ Therefore, for the 243 synthesis of α, ω -diaminopropyl PDMS, a base-type catalyst, such as for example 244 an anion-exchange resin, should be used.⁹ As a suitable anion exchange resin is 245 246 not available at the moment, it was decided to use TMAH, a homogeneous cat-247 alyst, but very easy to handle, which is not usual for this type. TMAH is unstable 248 at higher temperatures, and simple heating at 150 °C provokes its complete ther-249 mal degradation, which means there was no need for further neutralization and 250 purification of the reaction mixture. The only drawback of TMAH in comparison 251 with heterogeneous catalysts is the impossibility of its recycling and reuse.



254 The mole ratio of D₄ and disiloxane determines the degree of polymerizat-255 ion, *i.e.*, the molecular weight of linear telechelic PDMSs. The ratio of D_4 and 256 disiloxane was selected to obtain molecular weights of 2500 g mol⁻¹ in all syntheses (*i.e.*, degree of polymerization \cong 30). In the calculation of the initial 257 258 reaction mixture compositions, it was assumed that 90 wt. % of the starting silox-259 ane units, [Si(CH₃)₂O], originating from D₄, would be incorporated in the linear 260 polymer, considering that the usual proportion of cyclics in equilibrates is about 261 10 wt. %.^{1,2}

262 After completion of the equilibration reaction, the cyclics were separated 263 from the linear polymer by distillation under the reduced pressure, as described 264 in Experimental. The weight of the cyclics was calculated from the weight of 265 equilibrate before distillation and the weight of linear polymer after distillation. 266 The composition of the initial reaction mixtures and the compositions with res-267 pect to linear polymers and cyclics (D₄ and higher) after completion of the react-268 ions, i.e., at assumed equilibrium, are given in Table I. It could be seen that the 269 found fractions of cyclics at assumed equilibrium varied from 8.2 wt. % for the 270 sample PDMS-COOH to 10.6 wt. % for the sample PDMS-NH₂. The content of 271 cyclosiloxanes in the equilibrates was in agreement with literature data.¹⁻³ 272 Simultaneously, the calculated degrees of polymerization, $\overline{X}_{n(calc)}$, which are 273 given in Table I, varied from 29.6 (PDMS-COOH) to 30.4 (PDMS-NH₂).

The calculated degrees of polymerization and the molecular weights of the polymers were obtained by a procedure shown for the sample PDMS-CH₃. From the difference in the weight of D_4 and of the cyclics after distillation (2.8 g), the

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- weight of the siloxane units, $[(CH_3)_2SiO]$ incorporated into the linear chains were calculated: 28.05 g - 2.80 g = 25.25 g.
- 279 The number of moles of siloxane units was calculated from the proportion:

280 1 mol:74 g =
$$x:25.25$$
 g, $x = 0.3412$ mol

- 281 where the molecular weight of siloxane unit is 74 g mol⁻¹.
- 282 TABLE I. Composition of the initial and equilibrium mixtures and the calculated average 283 number degree of polymerization, $\overline{X}_{n(calc)}$ of α, ω -telechelic PDMSs

о. л. ·		Initial mixtures		Equilibrium mixtures		D 1	
Synthesis	Type of polymer		$m_{\rm D4}$ / g ^b	Weight fraction, %		Polymer	
INU.		$m_{\rm DS}$ / g.		Polymer	Cyclics	A n(calc)	
1	PDMS-CH ₃	1.85	28.05	90.6	9.4	29.9	
2	PDMS-H	1.44	28.50	90.2	9.8	29.8	
3	PDMS-CH=CH ₂	2.10	27.76	91.6	8.4	30.1	
4	PDMS-COOH	2.49	20.05	91.8	8.2	30.4	
5	PDMS-NH ₂	2.72	27.00	89.4	10.6	29.6	

^aMole fraction of DS was in all cases 0.11; ^bmole fraction of D_4 was in all cases 0.89

From the mole ratio of siloxane units that are incorporated in the linear polymer chains and the number of moles of HMDS (1.85 g \Rightarrow 0.0114 mol), the number average degree of polymerization, $\overline{X}_{n(calc)}$ was calculated:

288
$$\overline{X}_{n(calc)} = \frac{0.3412 \text{ mol}}{0.0114 \text{ mol}} = 29.9$$

289 The number average molecular weight, $\overline{M}_{n(calc)}$, was calculated based on 290 $\overline{X}_{n(calc)}$:

291
$$\overline{M}_{n(calc)} = 74 \overline{X}_{n(calc)} + 162 = 2380 \text{ g mol}^{-1}$$

where 162 g mol⁻¹ is the molecular weight of disiloxane HMDS.

The calculated molecular weights of the polymers, $\overline{M}_{n(calc)}$, are given in Table II. It could be seen that the values of $\overline{M}_{n(calc)}$ were in the range from 2340 to 2560 g mol⁻¹.

296 TABLE II. Molecular weights of α, ω -telechelic PDMSs

Polymer	$\overline{M}_{n(calc)}$ g mol ⁻¹	\overline{M}_{nNMR} g mol ⁻¹	\overline{M}_{nGPC} g mol ⁻¹	\overline{M}_{wGPC} g mol ⁻¹	Ι	M _{peak} g mol⁻¹	[η] dL g ⁻¹	\overline{M}_{η} g mol ⁻¹
PDMS-OH	2500-3000a	-	3420	5600	1.64	4000	_	-
PDMS-CH ₃	2380	-	2890	5420	1.88	4750	0.051	4510
PDMS-H	2340	8170	5240	9790	1.87	8460	0.062	6090
PDMS-CH=CH ₂	2410	5170	4430	7760	1.75	6630	0.055	5070
PDMS-COOH	2560	3100	3160	6280	1.99	4750	0.044	3590
PDMS-NH ₂	2440	3720	2820	7010	2.49	9920	0.061	5930

^aGiven in the producer's specifications

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The structures of the disiloxanes and D_4 , and of the assumed structures of the polymers, given in Fig. 2, were confirmed by ¹H-, ¹³C- and 2D-NMR spectroscopy. The spectra are depicted in Figs. S-2 (¹H-NMR), S-3 (¹³C-NMR) and S-4 (two-dimensional DEPT spectrum) of the Supplementary material.

302 The ¹H-NMR spectrum of PDMS-CH₃ (Fig. S-2 of the Supplementary 303 material) showed a signal at 0.06 ppm of the protons of the methyl groups bound 304 to silicon. The ¹H-NMR spectrum of PDMS-H (Fig. S-2) showed a signal at 0.14 305 ppm of the Si-CH₃ protons and a signal at 4.71 ppm of the protons connected 306 directly to silicon, Si-H. The signal at 0.07 ppm indicates the presence of mono-307 functional polymer, *i.e.*, arises from the methyl groups of 1,1,1-trimethyl-sub-308 stituted silicon atoms. GC of TMDS also confirmed the presence of two disilox-309 ane compounds. The contribution of TMDS in gas chromatogram was 82.7 % 310 while the contribution of 1,1,1,3,3-pentamethyl-disiloxane was 12.9 %. There 311 was also found a small fraction of other compounds (4.4 %).

The ¹H-NMR spectrum of PDMS-CH=CH₂ (Fig. S-2) showed a signal at 0.15 ppm of the protons from the methyl groups bound to silicon, as well as a multiplet at 5.95 ppm of the vinyl-group protons, CH=CH₂. The ¹³C-NMR spectrum of PDMS-CH=CH₂ (Fig. S-3) showed signals at 131.66 and 139.40 ppm of the CH and CH₂ carbons of the vinyl groups respectively, while the signal at 1.00 ppm arises from the carbons of the methyl groups connected to silicon.

318 The ¹H-NMR spectrum of PDMS-COOH (Fig. S-2) showed signals at 0.59, 319 1.69 and 2.38 ppm of the protons from the propane-1,3-diyl CH₂ groups (closer to the carboxyl group have higher δ shift), while at 0.05 ppm, the signal of Si-320 321 CH₃ protons is observed. In the spectrum of DCPTMDS disiloxane, the signal of the protons from the carboxyl-groups, COOH was found at 11.42 ppm. This 322 323 signal was not found in the spectrum of PDMS-COOH because of the low concentration of terminal COOH groups. The ¹³C-NMR spectrum of PDMS-324 325 COOH (Fig. S-3) showed signals at 17.84, 18.85 and 37.38 ppm of the carbon 326 atoms from CH₂ groups. The signal at 1.00 ppm is of the carbon atoms from the 327 methyl groups bound to silicon, while the signal at 179.72 ppm comes from the 328 carbon atoms of the carboxyl groups, COOH.

329 The ¹H-NMR spectrum of PDMS-NH₂ (Fig. S-2) showed a signal at 0.07 330 ppm of the protons of the methyl groups bound to silicon. The signals at 0.52, 331 1.50 and 2.66 ppm are from the CH₂ protons from the propane-1,3-diyl groups, while the signal at 1.38 ppm arises from NH_2 protons. The signal at 0.96 ppm is 332 333 from the protons of methyl groups bound to the methyne (CH) groups. Analysis 334 of the ¹H-NMR spectrum showed that the commercial DAPTMDS product is a 335 mixture of structural isomers (Fig. 4). As all three compounds participate in the equilib-336 ration reactions, the resulting PDMS-NH2 polymers may contain both propane-1,3-337 diyl (-CH₂-CH₂-CH₂-, Fig. 4-I) or both propylene (-CH(CH₃)CH₂-, Fig. 4-II) moieties, as well as one propane-1,3-diyl and one propylene group (Fig. 4-III). 338



The ¹³C-NMR spectrum of PDMS-NH₂ (Fig. S-3 of the Supplementary 341 342 material) showed a signal at 0.09 ppm of carbons from the methyl groups bound 343 to silicon. The signals at 15.14, 27.44 and 45.21 ppm arise from CH₂ carbons of 344 the propane-1,3-diyl groups. The signal of carbon from the methyl groups bound 345 to a methyne group appears at 11.36 ppm. The signal at 26.20 ppm arises from 346 methyne carbons, while the signal at 40.08 ppm is from CH₂ carbons connected 347 to amino groups. Based on the two-dimensional DEPT spectrum of DAPTMDS 348 (Fig. S-4 of the Supplementary material), it was concluded that four different 349 methyl groups exist showing signals at -1.47, -1.10, 0.10 and 11.37 ppm, as well 350 as three methylene (CH₂) groups at 15.14, 21.00, 27.45 and 45.21 ppm and one 351 methyne (CH) group at 26.21 ppm.

339 340

352 Gas chromatogram of DAPTMDS confirmed the presence of three types of 353 very similar molecular species. DAPTMDS participated with 61.8 %, while the 354 isopropyl structural isomers with one or two branched groups participated with 355 28.0 and 10.2 %.

The polymer sample PDMS-NH₂, and the mixture of cyclosiloxanes obtained 356 357 as by-products in the synthesis were characterized by IR spectroscopy. The IR 358 spectrum of PDMS-NH₂ showed characteristic peaks at 2963 and 2905 cm⁻¹ (asymmetric and symmetric C-H stretching), 1584 (NH₂ bending), 1412 (SiMe 359 asymmetric), 1261 (SiMe symmetric), 1083 and 1024 cm⁻¹ (SiOSi asymmetric 360 361 stretching, broad band with two absorption maxima) and 800 cm⁻¹ [Si(CH₃)₂ 362 rocking]. The bands of very low intensity at about 3500 cm⁻¹ arise from the asymmetric and symmetric stretching of NH₂ groups. The IR spectrum of the 363 364 cyclosiloxanes obtained after equilibrate distillation (Synthesis No. 5), showed

characteristic bands at 2963 and 2905 cm⁻¹ (asymmetric and symmetric C–H
stretching), 1412 and 1261 cm⁻¹ (SiMe asymmetric and SiMe symmetric), as
well as a broad band with one absorption maximum at 1074 cm⁻¹ (SiOSi asymmetric). Bands from NH₂ groups, about 3500 cm⁻¹, were not detected in this
spectrum. This was indirect confirmation that all the structural amino-disiloxane
isomers were completely incorporated in PDMS-NH₂.

371 Molecular weights of the α,ω-telechelic PDMSs

372 All the relevant molecular weight values of the α, ω -telechelic PDMSs, both 373 calculated and experimentally determined, are listed in Table II. The first column 374 gives the $M_{n(calc)}$, obtained from experimentally determined reaction mixture 375 compositions at equilibrium. The number-average molecular weights of the 376 obtained linear polymers were experimentally determined using different tech-377 niques. First, the $M_{\rm nNMR}$ values were calculated from the ¹H-NMR spectra, via 378 the experimentally obtained ratio of the signal integrals for the methyl group and 379 the end-group protons. Second, from GPC, which gave both the number-average, 380 M_{nGPC} , and the weight-average, M_{wGPC} , molecular weights and finally from 381 the intrinsic viscosities, $[\eta]$, which gave viscosity average values, $M \eta$.

382 The M_{nNMR} values of the samples PMDS-COOH and PDMS-NH₂ (3100 and 3720 g mol⁻¹, respectively) were in relatively good agreement with the <u>calculated</u> values, $\overline{M}_{n(calc)}$, (2560 and 2440 g mol⁻¹, respectively). The 383 384 385 M_{nNMR} values of the samples PMDS-CH=CH₂ and PDMS-H (5170 and 8170 g 386 mol⁻¹, respectively) were significantly higher in comparison with the $M_{n(calc)}$ 387 values (2410 and 2340 g mol⁻¹, respectively). One of the possible reasons for this 388 might be a significant difference in the intensity of the protons of a methyl group 389 and the protons of the end-groups. The higher $M_{\rm nNMR}$ value of the sample 390 PDMS-H could also be explained by lower purity of TMDS, *i.e.*, the presence of 391 1,1,1,3,3-pentamethyldisiloxane. Except for the terminal hydrido groups, that 392 also causes the presence of (trimethylsilyl)oxy groups and, consequently higher 393 intensity of the methyl group integral, leading to a false higher $M_{\rm nNMR}$ value of 394 PDMS-H. Furthermore, TMDS possesses the lowest boiling point of all the 395 disiloxanes used in this work, and hence, a deviation of M_{nNMR} from $M_{n(calc)}$ 396 could be the consequence of its higher volatility during the synthesis, which 397 again causes an increased ratio of Si(CH₃)₂ to end-group protons. The molecular 398 weight of the sample PDMS-CH₃ could not be determined by ¹H-NMR, since it 399 was very difficult to distinguish protons from the terminal Si(CH₃)₃ groups and 400 protons from Si(CH₃)₂ groups within the chain.

401 Gel-permeation chromatograms of the examined samples are given in Fig. 5. 402 The GPC measurements gave relative molecular weight values of the α, ω -tele-403 chelic PDMSs according to siloxane viscosity standards. Number and weight 404 average molecular weights (\overline{M}_{nGPC} and \overline{M}_{wGPC}) were obtained, as well as

405 polydispersity indices ($I = M_{wGPC}/M_{nGPC}$). The values corresponding to the top of the peaks, M_{peak} , were also determined. The results are given in Table II. 406 Matching between $M_{n(calc)}$ and \overline{M}_{nGPC} was good for the samples PDMS-CH₃, 407 408 PDMS-COOH and PDMS-NH₂. Deviation of M_{nGPC} from $M_{n(calc)}$ was again 409 the highest for the samples PDMS-H (5240 g mol⁻¹) and PDMS-CH=CH₂ (4430 410 g mol⁻¹), but still less in comparison with deviation of its M_{nNMR} values from 411 $\overline{M}_{n(calc)}$. It is not surprising that the experimental points obtained using ¹H-412 -NMR spectroscopy and GPC measurements are more or less scattered about the 413 expected (calculated) values but, in general, it seems that, in this particular case 414 and the examined molecular weights range, the GPC method might be the more 415 useful and reliable experimental technique for determination of the number 416 average molecular weights.



417 The results of the GPC measurements were used to calculate the polydis-418 persity indices I, a measure of the molecular weight distribution width and their 419 values are also given in the Table II. The polydispersity indices were in range from 1.75 (PDMS-CH=CH₂) to 2.49 (PDMS-NH₂), indicating the "most pro-420 421 bable" molecular weight distribution, characteristic for polymers obtained by step 422 polymerization (Fig. 5). It should be mentioned that the polydispersities obtained 423 here are in fair agreement with the results established for samples of comparable molecular weights.¹⁰ The highest I value of 2.49 was shown by PDMS-NH₂, 424 425 which might be explained by the presence of disiloxane isomers and their differ-426 ent reactivity, which leads to the spread of the molecular weight distribution.

427 Regarding the molecular weights, the α, ω -telechelic PDMSs were finally 428 characterized by dilute solution viscometry. The intrinsic viscosities were deter-429 mined, as described in the Experimental, and then the average viscosity values, 430 $\overline{M} \eta$, were calculated using the Mark–Houwink equation. The $\overline{M} \eta$ values were in 431 range from 3590 to 6090 g mol⁻¹ (Table II). The $\overline{M} \eta$ values were between

432 M_{nGPC} and M_{wGPC} (closer to M_{wGPC}) in all cases, as was expected for step 433 polymerizations.

434 Thermogravimetric analysis of α,ω-telechelic PDMSs

The influence of the type of end-groups on the thermal and thermo-oxidative stability of α, ω -telechelic PDMSs was examined by thermogravimetric analysis in nitrogen and air. Thermogravimetric (TG) and differential thermogravimetric (DTG) curves of the synthesized α, ω -telechelic PDMSs and the commercial samples – B3 ((trimethylsilyl)oxy-terminated PDMS) and PDMS-OH (silanolterminated PDMS) are presented in Fig. 6. The characteristic results are given in Tables III and IV.

442 The literature data suggest that during the thermogravimetric analysis, in the 443 absence of oxygen, polysiloxanes generally degrade by a single weight-loss step. The characteristic temperatures of this process - the onset of degradation, the 444 445 maximal rate of degradation, the end of degradation, etc., depend primarily on 446 the type of the polymer side groups and on the purity of the sample examined. On 447 the other hand, if the side groups in all samples are entirely CH₃, as in this study, 448 then the influence of the end-groups on the degradation mechanism could be 449 examined. Additionally, the relatively low molecular weights of the examined 450 polymers allow this effect to be easily registered.

From results obtained herein, it is obvious that the end-groups significantly affect the thermal and thermo-oxidative stability as well as the degradation mechanisms of α, ω -telechelic PDMSs. The temperatures at which the weight loss was 5 wt. % ($T_{5\%}$) were taken as indicators of the onset of degradation. In addition, the temperatures at which the weight loss was 50 wt. % ($T_{50\%}$), as well as the temperatures of the maximal rate of degradation (T_{max}) and corresponding residue upon degradation at 700 °C were taken as characteristic degradation parameters.

458 It could be seen that all α, ω -telechelic PDMSs started to lose weight at 459 somewhat lower temperatures in air (236-352 °C) than in nitrogen (265-360 °C). 460 The $T_{5\%}$ values were highest for the samples PDMS-H and PDMS-CH=CH₂ 461 both in nitrogen (360 and 332 °C, respectively) and air (352 and 314 °C, respect-462 ively) atmosphere, which was somehow surprising result. The possible cause of 463 this behavior might be the initial addition reactions, which are possible thanks to 464 the presence of Si-H or Si-CH=CH₂ groups, without the generation of volatile 465 products. If no volatile compounds are formed in PDMS-H and PDMS-CH=CH₂, 466 then there is no weight loss below and about 300 °C, when the other samples 467 started to degrade.

468 The commercial samples B3 and PDMS-OH (terminated with SiCH₃ and 469 Si–OH groups, respectively) start to degrade at very similar temperatures in both 470 nitrogen and air atmospheres (Tables IV and V). However, the degradation mech-471 anisms were completely different, which could be seen from the shape of their DTG



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Comment [N6]: Too small font in Fig. 6. Shou be rearranged.

472 TABLE III. The data of the thermogravimetric analysis of the α, ω -telechelic PDMSs in nitro	ogen
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Sample	$T_{5\%} / {}^{\circ}\mathrm{C}$	<i>T</i> _{50%} / °C	$T_{\rm max.}$ / °C	Residue at 700 °C, wt. %
PDMS-OH	312	469	320/456/582	0
B3	316	482	528	1.5
PDMS-CH ₃	266	478	526	0.3
PDMS-H	360	522	440/478/526/558	0.6
PDMS-CH=CH ₂	332	583	420/556/578/608	2.2
PDMS-COOH	265	507	278/430/538	4.0
PDMS-NH ₂	268	442	386/440	0.8

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473 7	ABLE IV. The data of the thermogravimetric analysis of the α, ω -telechelic PDMSs in air	
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Sample	$T_{5\%} / {}^{\circ}{ m C}$	$T_{50\%} / {}^{\circ}\mathrm{C}$	$T_{\rm max.}$ / °C	Residue at 700 °C, wt. %
PDMS-OH	306	516	356/464/500	45.2
B3	312	543	354/392/424/508	49.0
PDMS-CH ₃	266	536	352/388/432/510	48.2
PDMS-H	352	517	356/392/436/502	43.3
PDMS-CH=CH ₂	314	508	320/356/398/494	42.9
PDMS-COOH	257	401	270/414	10.1
PDMS-NH ₂	236	483	468/492	16.9

474 curves (Fig. 6). It is known that PDMS with SiOH end-groups degrades by
475 depolymerization in an inert atmosphere.²⁷ Thanks to the great flexibility of
476 -[SiO]*x*- segments of the siloxane chains, the terminal SiOH groups can initiate
477 intramolecular rearrangement by the so-called "back-biting" or "unzipping"

478 process, which results in the formation of cyclic siloxanes (Fig. 7).





Fig. 7. Thermal degradation of silanol-terminated PDMS – "unzipping" mechanism.

481 The similar shape of TG and DTG curves of the PDMS-CH₃ and the com-482 mercial B3 of analogous molecular weight was expected, as both samples were 483 terminated entirely by methyl groups. The lower $T_{5\%}$ value of the sample PDMS-484 -CH₃ (266 °C for both N₂ and air atmosphere), in comparison with $T_{5\%}$ value of 485 B3 (316 °C in N₂ and 312 °C in air) could be attributed to the presence of linear 486 siloxanes of lower molecular weights and also maybe the presence of residual 487 cyclics, that are not completely removed during the vacuum distillation. The

488 initial weight loss of the PDMS-CH₂ could be caused by the evaporation of lower 489 linear and cyclic siloxanes during the thermogravimetric analysis. As a precon-490 dition for intramolecular depolymerization in an inert atmosphere, which begins 491 from the chain-ends, is the presence of SiOH groups, it could be concluded that 492 PDMSs terminated with different kinds of functional groups cannot degrade in 493 this way. It is well known that the stability of the Si-O bonds is countered by the 494 characteristic flexibility of the siloxane chain segments, which enables easy 495 formation of local configurations that facilitate inter- and intramolecular siloxane redistribution reactions.²³ These reactions lead to fragmentation of linear silox-496 497 ane chains into the smaller cyclosiloxanes that are volatile and thermodinamically more stable at the degradation temperatures. Consequently, the stability of 498 499 linear PDMSs at elevated temperatures is reduced, and their degradation occurs 500 at temperatures lower than would be expected based on Si-O dissociation 501 energies.2,3

502 PDMS-CH₃ and B3 samples terminated with CH₃ groups may degrade by 503 inter- and intramolecular redistribution reactions, which occur randomly between 504 the siloxane bonds located within the polymeric chains, whereby cyclics re-505 occur. This is the so-called "random scission" mechanism, shown in Fig. 8.^{29,34}



506 507 508

Fig. 8. Thermal degradation of PDMS – "random scission" mechanism; A) intramolecular redistribution and B) intermolecular redistribution.

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The samples PDMS-COOH and PDMS-NH₂ were the most thermally unstable in both N₂ and air atmospheres, in comparison with the other telechelic PDMSs. The $T_{5\%}$ values were 265 °C for the sample PDMS-COOH and 268 °C for PDMS-NH₂ in nitrogen, whereas the corresponding temperatures in air amounted to 257 °C (PDMS-COOH) and 236 °C (PDMS-NH₂). The degradation mechanisms of these two samples are completely different in both nitrogen and air atmosphere, which was confirmed by the DTG curves shown in Fig. 6.

The degradation of PDMS-COOH in N_2 occurs in three steps, of which the first one is probably decarboxylation, followed by unzipping and/or random scission. It can be seen that the same sample in air degrades by a two-stage mechanism.

520 Based on the DTG curves, it could be concluded that majority of the exam-521 ined α, ω -telechelic PDMSs degrade according to complex mechanism, with the 522 maximal weight loss rates occurring in at least two temperatures intervals. The 523 sample PDMS-OH shows two-step degradation processes, with T_{max} at 456 and 524 582 °C. A "knee" at the TG curve of the PDMS-OH sample can be clearly seen, 525 which signifies where the first degradation step ends and the second step begins. 526 Most probably, the degradation of PDMS-OH in N₂ begins from the chain-ends by 527 "back-biting" (the first step of degradation) and then proceeds by "random scis-528 sion" (the second step of degradation). The exception is the degradation of the 529 samples B3 and PDMS-CH3 in N2, where one maximum on the DTG curves 530 occurs in both cases (528 and 526 °C, respectively). The one-step degradation process of these samples involves only "random scission". A $T_{\rm max}$, corresponding to 531 532 "random scission" also occurs in the DTG curves of the samples PDMS-H, PDMS-533 -CH=CH₂ and PDMS-COOH at 558, 601 and 538 °C, respectively (Table III).

534 In contrast to thermal degradation in N2, the literature data suggest that thermo-oxidative degradation of the same PDMSs is generally a more complex 535 process, which leads to pure silica at temperatures above 600 °C. The data obta-536 537 ined here confirmed this suggestion in the case of the samples PDMS-CH₃ and 538 B3, with terminal methyl groups, where four T_{max} can be seen at each DTG curve. The thermo-oxidative degradation of the other samples proceeds through 539 540 at least two steps (Fig. 6, left panel), but compared to the thermal degradation of 541 the same samples in nitrogen, their mechanisms cannot be denoted as more 542 complex.

The residue after degradation was very low or even it was not present at all in N₂ (from 0 to 4 wt.%, Table III), while in air, the amount of residue ranged from 10.1 wt. % (PDMS-COOH) to 49.0 wt.% (B3) (Table IV). The explanation for this behavior can be found again in different degradation mechanisms of PDMS chains in inert and oxidative atmosphere. In inert N₂ atmosphere, regardless of whether degradation occurs by unzipping or random scission, cyclosiloxanes as thermally more stable products in comparison with linear PDMSs are



550 formed, and because of this, the residual weights were very small (Table III). In a 551 thermo-oxidative atmosphere, degradation of the PDMS chains starts with form-552 ation of peroxy structures in the side-chains, which leads to the occurrence of 553 different free radical reactions and results in the formation of CO, CO₂, H₂, H₂O, 554 HCHO and HCOOH, leaving a significant amount of residual SiO2 above 600 °C 555 (Fig. 9). The amount of the resulting residue in air was usually smaller than what 556 would be expected from the relative content of silicon in the original polymer, meaning that volatile organosilicon compounds were also formed.^{3,23,24} 557



558 559

Fig. 9. Thermo-oxidative degradation of PDMS by the free-radical mechanism.

560 Taking into account all the observations described for the thermo-oxidative degradation of PDMSs, the free-radical mechanism is proposed until 350-400 °C, 561 followed by siloxane redistribution reactions at higher temperatures.³⁵ A slightly 562 surprising result was the relatively small residue that was found in the samples 563 PDMS-COOH and PDMS-NH2 (10.1 and 16.9 wt. %, Table IV). This indicates 564 565 that during oxidative degradation of PDMS-COOH and PDMS-NH2, a large quantity of volatile organosilicon compounds were produced in comparison with other 566 567 examined samples, where the residues in air were between 40 and 50 wt. %.

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CONCLUSIONS

569 A series of α, ω -telechelic PDMS with hydrido, vinyl, carboxypropyl and 570 aminopropyl functional groups at the polymer chain-ends, with target molecular 571 weight of about 2500 g mol⁻¹, were prepared by siloxane equilibration reactions. 572 The number average molecular weights of the samples PMDS-COOH and PDMS-NH₂, calculated from their ¹H-NMR spectra (3100 and 3720 g mol⁻¹, res-573 574 pectively), agreed very well with the values predetermined by the compositions 575 of the reaction mixtures. The M_{nNMR} values of the samples PMDS-CH=CH₂ 576 and PDMS-H (5170 and 8170 g mol⁻¹, respectively), were significantly higher in 577 comparison with the calculated values (2410 and 2340 g mol⁻¹, respectively). 578 Except of the meaningful difference in the intensities of the proton signals from 579 terminal groups, and the protons from the methyl groups, the reasons for the 580 higher $M_{\rm nNMR}$ value of the PDMS-H sample could be the lower purity and 581 higher volatility of the TMDS reactant, which has the lowest boiling temperature 582 of all the disiloxanes used in this work. Regarding the size of polymer molecules, 583 the synthesized polymers were also characterized by GPC and dilute solution 584 viscometry. Agreement between $\overline{M}_{n(calc)}$ and \overline{M}_{nGPC} was very good for the 585 samples PDMS-CH₃, PDMS-COOH and PDMS-NH₂. The maximal deviation between M_{nGPC} and $M_{n(calc)}$ was again for the sample PDMS-H. The polydis-586 587 persity index of the synthesized samples ranged from 1.75 (PDMS-CH=CH₂) to 588 2.49 (PDMS-NH₂), which suggested the "most probable" molecular weights dis-589 tribution. The $M \eta$ values obtained from dilute solutions viscometry lay between 590 M_{nGPC} and M_{wGPC} , as was to be expected for the "most probable" distribution. 591 The results of thermogravimetric analysis in nitrogen and air showed that the 592 terminal functional groups influenced significantly the thermal and thermo-oxi-593 dative stability of the α, ω -telechelic PDMSs, as well as the degradation mech-594 anism. Degradation of all polymer samples started at lower temperatures in an 595 oxidative atmosphere than in an inert nitrogen atmosphere. The highest tempe-596 ratures of the onset of degradation were shown by the samples PDMS-H and 597 PDMS-CH=CH₂, in both a nitrogen and an air atmosphere. It is believed that this 598 behavior pattern of the PDMS-H and PDMS-CH=CH₂ is the consequence of 599 initial addition reactions, without the generation of volatile products at tempe-600 ratures below and about 300 °C, which were possible thanks to the presence of 601 the Si-H or Si-CH=CH₂ groups. The samples PDMS-COOH and PDMS-NH₂ 602 were the most unstable in both nitrogen and air atmosphere in comparison with 603 the other investigated telechelic PDMSs. The shape of the DTG curves having at 604 least two weight loss peaks showed that the majority of the α, ω -telechelic 605 PDMSs degrade by complex mechanisms. The residue after degradation was 606 very low in nitrogen (0 to 4 wt. %), while in the air, it was significantly higher 607 (10 to 49 wt. %). In order to obtain more details about the degradation products 608 and degradation mechanism of α, ω -telechelic PDMSs, thermogravimetric ana-

609 610	lysis with mass detection (TG–MS), as well as pyrolysis-gas chromatography- mass spectrometry analysis (Py–GC–MS) are planned for further research.
611	SUPPLEMENTARY MATERIAL
612 613 614 615	The data of PDMS Brookfield standards and GPC calibration curve, as well as ¹ H-, ¹³ C- and DEPT-NMR spectra are available electronically at the pages of journal website: http://www.shd.org.rs/JSCS/, or from the corresponding author on request.
616 617	Acknowledgment. This work was financially supported by the Ministry of Education, Science and Technological Development of the Republic of Serbia (Project No. 172062).
618 619 620	извод СИНТЕЗА, СТРУКТУРА И ТЕРМОГРАВИМЕТРИЈСКА АНАЛИЗА а, Ф-ТЕЛЕХЕЛИЧНИХ ПОЛИДИМЕТИЛСИЛОКСАНА МАЛЕ МОЛСКЕ МАСЕ
621	АЛЕКСАНДРА М.ТАСИЋ ¹ , МАРИЈА В. ПЕРГАЛ ² , МАЛИША П. АНТИЋ ³ и ВЕСНА В. АНТИЋ ³
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$\begin{array}{c} 625\\ 626\\ 627\\ 628\\ 629\\ 630\\ 631\\ 632\\ 633\\ 634\\ 635\\ 636\\ 637\\ 638\\ 639 \end{array}$	Серија α, ω -телехеличних полидиметилсилоксана (PDMS), предодређене молске масе од око 2500 g mol ⁻¹ , синтетисана је реакцијом силоксанске еквилибрације. Синтезе су изведене полазећи од октаметилциклотетрасилоксана (D ₄) и различитих дисилоксана, као што су: хексаметилдисилоксан (HMDS), 1,1,3,3-тетраметилдисилоксан (TMDS), 1,3-дивинилтетраметилдисилоксан (DVTMDS), 1,3-бис(3-карбоксипропил)тетраметилдисилоксан (DVTMDS), 1,3-бис(3-карбоксипропил)тетраметилдисилоксан (DVTMDS). Дисилоксани су служили за увођење функционалних група на крајеве полимерних ланаца, а такође су имали улогу регулатора молске масе полимера. Полидиметилсилоксани са триметил, хидридо, винил, карбоксипропил и аминопропил завршним групама су добијени на овај начин. Структура α, ω -телехеличних PDMS је потврђена NMR и IR спектроскопијом. Молске масе полимера су одређене ¹ H-NMR спектроскопијом, гел-пропусном хроматографијом (GPC) и вискозиметријом разблажених раствора. Термогравиметријска анализа (TGA) у азоту и ваздуху је показала да природа завршних група значајно утиче на термичку и термо-оксидативну стабилност, као и на механизам деградације α, ω -телехеличних PDMS.
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641	REFERENCES
642 643 644 645	 I. Yilgor, J. E. McGrath, Adv. Polym. Sci. 86 (1988) 1 M. G. Voronkov, V. P. Mileshkevich, Yu. A. Yuzheleviskii, The Siloxane Bond, Consultants Bureau, New York, 1978 P. P. Dvornić, P. W. Lenz, High Temperature Siloxane Elastometry, Huthing and Wort
646	Verlag, Basel, 1991

- 647
- K. Kojima, C. R. Gore, C. S. Marvel, J. Polym. Sci., Part A-1: Polym. Chem. 4 (1966) 2326
 J. S. Riffle, I. Yilgor, C. Tran, G. L. Wilkes, J. E. McGrath, A. K. Banthia, in Epoxy Resin Chemistry, R. S. Bauer, Ed., ACS Symposium Series, Vol. 221, ACS, Washington DC, 648 649 650 1983
- 651 6. A. Ekin, D. C. Webster, J. Polym. Sci., Part A: Polym. Chem. 44 (2006) 4880

652

TASIĆ et al.

653	8. P. R. Dvornić, M. N. Govedarica, J. D. Jovanović, V. V. Gerov, M. P. Antić, Polym. Bull.
654	35 (1995) 539
655	9. M. Cazacu, M. Marcu, A. Vlad, D. Caraiman, C. Racles, Eur. Polym. J. 35 (1999) 1629
656	10. V. M. Đinović, V. V. Antić, J. Đonlagić, M. N. Govedarica, React. Funct. Polym. 44
657	(2000) 299
658	11. P. M. Antić, M. V. Vučković, V. V. Antić, M. N. Govedarica, Mater. Sci. Forum 453-
659	-454 (2004) 199

7. M. N. Govedarica, P. R. Dvornić, J. Serb. Chem. Soc. 56 (1991) 7

- 660 12. A. Gehrig, K. Humburg, K. T. Nguyen, M. D. Telgenhoff, R. Wells (Dow Corning 661 Corporation, Midland, MI, USA), US Patent 8 686 175 (2014)
- 13. E. R. Evans (General Electric Company, Waterford, NY, USA), US Patent 4 122 247 662 663 (1978)
- 664 14. U. Finke, H. H. Moretto (Bayer Aktiengesellschaft, DE) US Patent 4 310 679 (1982)
- 665 15. B. A. Janeiro (Gelest, Inc., Morrisville, PA, USA), US Patent 0 176 561 A1 (2004)
- 666 16. J. D. Jovanović, M. N. Govedarica, P. R. Dvornić, I. G. Popović, Polym. Degrad. Stab. 61 667 (1998) 87
- 668 17. S. Andre, F. Guida-Pietrasanta, A. Rousseau, B. Boutevin, G. Caporiccio, Polymer 42 669 (2001) 5505
- 670 18. G. Camino, S. M. Lomakin, M. Lazzari, Polymer 42 (2001) 2395
- 671 19. G. Camino, S. M. Lomakin, M. Lageard, Polymer 43 (2002) 2011
- 672 20. G. Deshpande, M. E. Rezac, Polym. Degrad. Stab. 76 (2002) 17
- 673 21. S. Zhang, H. Wang, J. Therm. Anal. Calorim. 103 (2011) 711
- 674 22. M. A. Brook, Silicon in Organic, Organometallic, and Polymer Chemistry, Wiley, New 675 York. 2000
- 676 23. S. J. Clarson, in Siloxane Polymers, S. J. Clarson, J. A. Semlyen, Eds., Polym. Sci. Techn. Series, Elis Horwood-PTR Prentice Hall, Englewood Cliffs, NJ, 1993 677
- 24. P. R. Dvornic, in Silicon-Containing Polymers, R. G. Jones, W. Ando, J. Chojnowski, 678 679 Eds., Kluwer Academic Publishers, Dordrecht, 2000
- 680 25. P. R. Dvornić, J. D. Jovanović, M. N. Govedarica, J. Appl. Polym. Sci., 49 (1993) 1497
- 681 26. A. C. M. Kuo, in Polymer Data Handbook, J. E. Mark, Ed., Oxford University Press, 682 Oxford, 1999
- 683 27. Yu. A. Aleksandrova, J. S. Nikitina, A. N. Pravednikov, Vysokomol. Soedin., A 10 (1968) 684 1078
- 685 28. V. V. Rode, V. A. Verkhotin, S. R. Rafikov, Vysokomol. Soedin., Ser. A 11 (1969) 1529
- 686 29. T. H. Thomas, T. C. Kendrick, J. Polym. Sci., Part A-2 7 (1969) 537
- 687 30. N. Grassie, I. G. Macfarlane, Eur. Polym. J. 14 (1978) 875
- 688 31. A. J. Barry, J. Appl. Phys. 17 (1946) 1020
- 689 32. M. Kurata, Y. Tsunashima, Polymer Handbook, 3rd ed., J. Brandrup, E. H. Immergut, Eds., Wiley, New York, 1989 690
- 691 33. C. L. Willis (Kraton Polymers, U.S. LLC, Houston, TX, USA), US Patent 8 263 713 B2 692 (2012)
- 693 34. T. H. Thomas, T. C. Kendrick, J. Polym. Sci., Part A-2 8 (1970) 1823
- 694 35. K. A. Andrianov, Metalorganic Polymers, Interscience, New York, 1965.

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695 SUPPLEMENTARY MATERIAL TO 696 Synthesis, structure and thermogravimetric analysis of α, ω -telechelic polydimethylsiloxanes of low molecular weight 697 ALEKSANDRA M. TASIĆ¹, MARIJA V. PERGAL², MALIŠA P. ANTIĆ³ and 698

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705 TABLE S-I. Bulk viscosities and calculated molecular weights of PDMS standards



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